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# A novel CoOOH/(Ti, C)-Fe<sub>2</sub>O<sub>3</sub> nanorod photoanode for photoelectrochemical water splitting

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**ABSTRACT** In this work, we demonstrate the CoOOH/(Ti, C)-Fe<sub>2</sub>O<sub>3</sub> (CTCF) nanorods prepared by a facile approach as well as their implementation as photoanodes for photoelectrochemical (PEC) water splitting. The photocurrent density of CTCF photoanode is 1.85 mA cm<sup>-2</sup> at +1.23 V vs. reversible hydrogen electrode (RHE), which is more than 20 times higher than that of pristine  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photoanode (0.08 mA cm<sup>-2</sup>). The incident-photo-to-current conversion efficiency, applied bias photo-to-current efficiency and transfer efficiency of CTCF photoanode reaches 31.2% at 380 nm (+1.23 V vs. RHE), 0.11% (+1.11 V vs. RHE), 68.2% (+1.23 V vs. RHE) respectively, which are much higher than those of pristine  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photoanode. Additionally, the longtime irradiation PEC water splitting of CTCF photoanode demonstrates its high stability at extreme voltage in NaOH (pH 14).

**Keywords:** photoelectrochemistry, water splitting, doping, ferric oxide

## INTRODUCTION

Solar to hydrogen energy conversion is one of the most promising future energy forms to solve the energy shortage and environmental problems, due to its renewable, carbon-free, and environmentally friendly features [1–3]. Different from the photocatalysis of water splitting, photoelectrochemical cell (PEC) hydrogen production has some outstanding advantages, such as higher solar utilization, easier to recycle and faster hydrogen production rate [1,4,5]. The photoelectrodes are the key factors to the efficiency of PEC water splitting like the leaves of photosynthesis in the nature [4]. Many efforts

have been devoted to exploring highly active photoanode materials such as TiO<sub>2</sub> [6,7], ZnO [8,9], Nb<sub>2</sub>O<sub>5</sub> [10], BiVO<sub>4</sub> [11,12] and WO<sub>3</sub> [13–15].

Hematite ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>) is one of the most promising earth-abundant catalysts for photoanode, which can absorb a substantial portion of the solar spectrum (300–600 nm) [16–19]. This feature makes  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> possess a remarkable maximum water oxidation photocurrent density ( $J_{\max}$ ) for solar water splitting [20]. However, some intrinsic features of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> in charge separation, electrical conductivity, material stability and surface oxygen evolution catalytic activity limit the PEC water splitting efficiency [21,22].

The  $J_{\max}$  of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> is more than 12 mA cm<sup>-2</sup>, but the reported water oxidation photocurrent densities of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> are within the range of 1–5 mA cm<sup>-2</sup> [20,23], due to the poor electrical conductivity [3,16,24], which results in inefficient migration rate and poor separation of photo-generated electrons and holes [25], resulting in solar energy degradation [22,23]. We believe that future enhancement of the PEC performance of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photoanode will focus on improving the electrical conductivity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>.

Doping is an effective method to enhance the electrical conductivity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>-based photoanode for PEC water splitting [24]. For example, Li *et al.* [16] indicated that a novel Sn-Fe<sub>2</sub>O<sub>3</sub> photoanode achieved the excellent photocurrent density of ~1.36 mA cm<sup>-2</sup> at 0.23 V vs. Ag/AgCl. The Sn element doping can effectively enhance its electrical conductivity. Recently, two or more elements doping of photoanodes are also catching researchers'

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wide attention [26]. Tang *et al.* [24] indicated that a novel  $\text{Fe}_2\text{O}_3$  with Ti, Sn and In doping photoanode achieved the excellent photocurrent density of  $\sim 2.2 \text{ mA cm}^{-2}$  at  $+1.23 \text{ V vs. Ag/AgCl}$ .

On the other hand, when the photo-generated electrons and holes are separated from the inside of  $\alpha\text{-Fe}_2\text{O}_3$ , the photo-generated electrons will move to the cathode of PEC cell for hydrogen evolution reaction (HER) and the photo-generated hole will move to the surface of photoanode for oxygen evolution reaction (OER) [1,23]. However, the PEC water splitting performance of  $\alpha\text{-Fe}_2\text{O}_3$  is limited because a majority of the holes are recombined when migrating to surface, leading to the poor catalytic activity of the  $\alpha\text{-Fe}_2\text{O}_3$  surface for oxygen evolution [24]. Recently, FeOOH as an oxygen evolution catalyst (OEC) attracted great attention due to its excellent OER catalytic capacity [24]. Similar to FeOOH, CoOOH is also an excellent OEC for electrical water splitting, but few studies were carried out on its application for PEC water splitting [27].

In this work, we utilize C and Ti doping  $\text{Fe}_2\text{O}_3$  to increase its electrical conductivity, and CoOOH OEC layer to enhance its OER capacity as an OEC material. The obtained CoOOH/(Ti, C)- $\text{Fe}_2\text{O}_3$  photoanode achieves  $1.85 \text{ mA cm}^{-2}$  at  $+1.23 \text{ V vs. RHE}$ , which is more than 20 times higher than that of pristine  $\alpha\text{-Fe}_2\text{O}_3$  photoanode ( $0.08 \text{ mA cm}^{-2}$ ).

## EXPERIMENTAL SECTION

### Synthesis of the catalysts

The  $\alpha\text{-Fe}_2\text{O}_3$  nanorod arrays were fabricated according to the literature [28]. 2.0 g ferric chloride ( $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$ ) was dissolved into 50 mL deionized water, and sealed in a Teflon-liner. The fluorine doped tin oxide (FTO)-coated glass of  $4 \text{ cm} \times 5 \text{ cm}$  was cleaned with deionized water, ethanol and acetone, and then placed in the liner with the FTO side facing the wall of the liner. The Teflon-liner was put into a self-sealing autoclave and heated at  $100^\circ\text{C}$  for 3 h. After the reaction, the yellow films formed on the FTO substrates were thoroughly rinsed with deionized water and transferred to a furnace. Then the yellow films coated substrates were annealed at  $550^\circ\text{C}$  for 2 h and then at  $750^\circ\text{C}$  for 30 min with a ramp rate of  $5^\circ\text{C min}^{-1}$  to obtain the  $\alpha\text{-Fe}_2\text{O}_3$ . The Ti- $\text{Fe}_2\text{O}_3$  nanorod arrays were prepared by adding 10  $\mu\text{L}$   $\text{TiCl}_3$  into the  $\text{FeCl}_3$  aqueous solution. The C- $\text{Fe}_2\text{O}_3$  nanorod arrays were obtained by adding 0.5 g glucose into the  $\text{FeCl}_3$  aqueous solution. The (Ti, C)- $\text{Fe}_2\text{O}_3$  nanorod arrays were obtained by adding 0.5 g glucose and 10  $\mu\text{L}$   $\text{TiCl}_3$  into the  $\text{FeCl}_3$  aqueous

solution. The (Ti, C)- $\text{Fe}_2\text{O}_3$  nanorod arrays was immersed in  $0.01 \text{ mol L}^{-1}$  cobalt acetate aqueous solution and polarized at 0 V vs. Ag/AgCl in dark using Pt mesh as counter electrode. Then it was illuminated with a sun light simulator with incident light power controlled to be  $100 \text{ mW cm}^{-2}$  and maintained for only 30 s to obtain the CoOOH/(Ti, C)- $\text{Fe}_2\text{O}_3$  nanorod arrays.

### Characterization

All the catalysts were characterized by field emission scanning electron microscope (SEM, JSM-6330F), X-ray diffractometer (XRD, D8 ADVANCE), X-ray photoelectron spectroscopy (XPS, ESCALab250), transmission electron microscope (TEM, JEM2010-HR), atomic force microscope (AFM, SPM-9500J3 microscope) and UV-vis spectrophotometer (UV-2450). The electrochemical testing was carried out with a CHI660C electrochemical station in a standard three electrode configuration.

### PEC measurements

Photoelectrochemical measurements were performed with an electrochemical analyzer in a three-electrode configuration using the catalysts on FTO as the working electrode, Pt wire as the counter electrode, and an Ag/AgCl reference electrode. An aqueous solution of  $1 \text{ mol L}^{-1}$  NaOH was used as electrolyte (pH 14). The working electrodes were immersed into the electrolyte and  $1 \text{ cm}^2$  surface areas exposed. The illumination source was AM 1.5G solar simulator (Newport, LCS 100 94011A (class A, Fig. S1)) directed at the quartz PEC cell ( $100 \text{ mW cm}^{-2}$ ). Monochromatic incident photon-to-electron conversion efficiency (IPCE) were collected by a Solartron 1280B electrochemical station with a solar simulator (Newport 69920, 1000 W xenon lamp), coupled with an infrared water filter (Oriel 6127) and aligned monochromator (Oriel Cornerstone 130 1/8 m). All the electrochemical measurements were performed on an SP-150 electrochemical workstation (SP-150, Bio-Logic SAS, France) at room temperature.

### Calculations

The conversion between potentials vs. Ag/AgCl and those vs. RHE is performed using the following equation [29]:

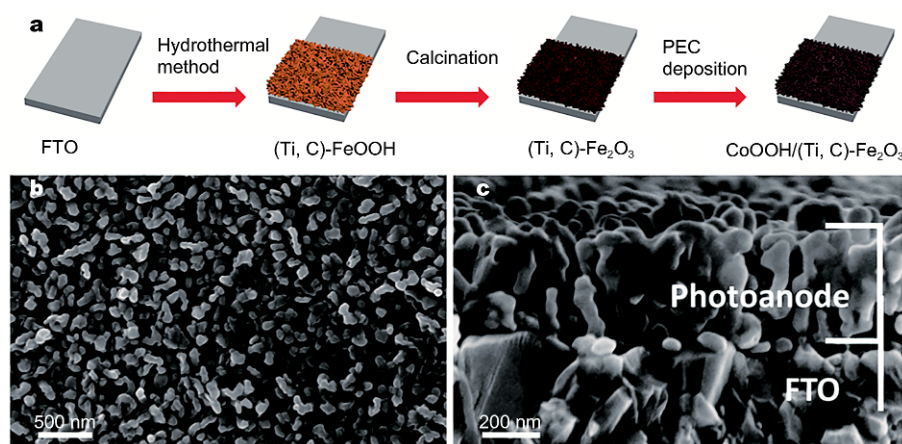
$$E \text{ (vs. RHE)} = E \text{ (vs. Ag/AgCl)} + E_{\text{Ag/AgCl}} \text{ (reference)} + 0.0591 \text{ V} \times \text{pH}$$

$$(E_{\text{Ag/AgCl}} \text{ (reference)} = 0.1976 \text{ V vs. NHE at } 25^\circ\text{C}).$$

IPCE can be expressed as [16]:

$$\text{IPCE} = (1240 \times I) / (\lambda \times J_{\text{light}}),$$

where  $I$  is the photocurrent density,  $\lambda$  is the incident light wavelength, and  $J_{\text{light}}$  is the measured irradiance.



**Figure 1** (a) Schematic diagrams for the growth of CoOOH/(Ti, C)-Fe<sub>2</sub>O<sub>3</sub> nanorods photoanode. (b) Top-view and (c) side-view SEM images of CoOOH/(Ti, C)-Fe<sub>2</sub>O<sub>3</sub> nanorods photoanode.

Yield of the surface reaching holes or the named transfer efficiency ( $\Phi_{\text{OX}}$ ) can be expressed as [29]:

$$\Phi_{\text{OX}} = J_{\text{PEC}} / J_{\text{HS}}$$

where  $J_{\text{PEC}}$  is the practical water oxidation photocurrent and  $J_{\text{HS}}$  is the photocurrent density of sample with Na<sub>2</sub>SO<sub>3</sub> hole scavenger.

Applied bias photo-to-current efficiency (ABPE) can be expressed as [12]:

$$\text{ABPE} = [J_{\text{PEC}} \times (1.23 - V_{\text{app}})] / P_{\text{light}}$$

where  $J_{\text{PEC}}$  is the photocurrent density of samples,  $V_{\text{app}}$  is the applied external potential vs. RHE and  $P_{\text{light}}$  is the power density of the illumination (100 mW cm<sup>-2</sup>).

## RESULTS AND DISCUSSION

As shown in Fig. 1a, the CoOOH/(Ti,C)-Fe<sub>2</sub>O<sub>3</sub> (CTCF) nanorods photoanode was synthesized on FTO glass substrate by three-step (Experimental section). Firstly, (Ti, C)-FeOOH nanorods film was synthesized hydrothermally. Then, the (Ti, C)-FeOOH nanorods film was transformed into (Ti, C)-Fe<sub>2</sub>O<sub>3</sub> by calcination. Finally, the CoOOH as an oxygen evolution catalyst was deposited onto the (Ti, C)-Fe<sub>2</sub>O<sub>3</sub> by a PEC deposition method.

The CTCF nanorods compactly grow on the FTO glass substrate with 400 nm long and 20 nm in diameter, as shown in the top-view SEM of CTCF (Fig. 1b) and the side-view SEM (Fig. 1c). XRD spectra were conducted to identify the crystalline phase of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, C-Fe<sub>2</sub>O<sub>3</sub>, (Ti, C)-Fe<sub>2</sub>O<sub>3</sub> (TCF) and CTCF, as shown in Fig. S2. Except the SnO<sub>2</sub> peaks originating from the FTO substrate, all other diffraction peaks in the XRD spectra can be well indexed to  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (JCPDS 86-0550).

Diffuse reflectance UV-visible spectra of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, C-Fe<sub>2</sub>O<sub>3</sub>, TCF and CTCF were collected to characterize the

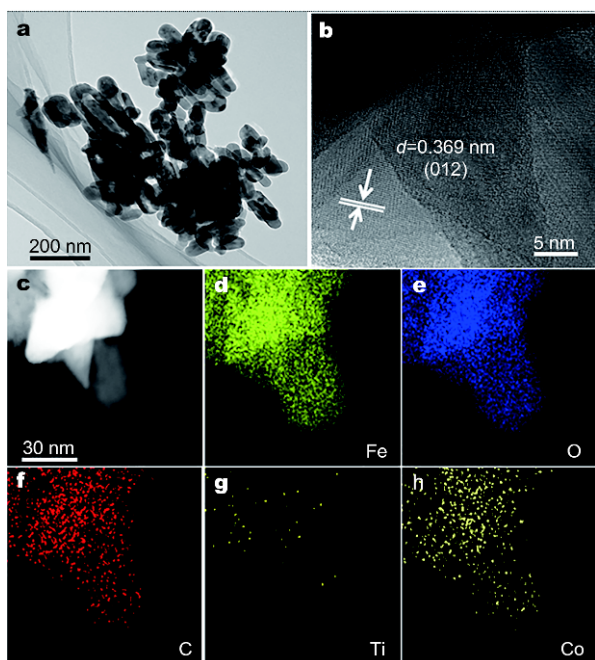
light harvesting capability and band gap. As shown in Fig. S3, the absorption edge of the as-synthesized samples are at 590 nm, which is consistent with recent reports. The band gap of samples can be determined by the formula [30]:

$$\alpha h\nu = A(h\nu - E_g)^{n/2}$$

where  $\alpha$ ,  $h$ ,  $\nu$ ,  $A$ ,  $E_g$  and  $n$  are absorption coefficient, Planck's constant, incident light frequency, proportionality constant, band-gap and characteristic integer, respectively. Among them,  $n$  depends on the characteristics of the optical transition in a semiconductor, i.e., direct transition ( $n = 1$ ) or indirect transition ( $n = 4$ ) [23]. Fe<sub>2</sub>O<sub>3</sub> pertains to indirect transition and the value of  $n$  is 4. As shown in Fig. S4, the absorption edges of the as-synthesized photoanodes are in the range of 590 to 600 nm, respectively, and the band-gap energies of the samples are 1.9 to 2.0 eV, respectively, which are consistent with recent reports [20,22,23].

Fig. S5 shows the XPS survey spectra of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, (Ti, C)-Fe<sub>2</sub>O<sub>3</sub> and CoOOH/(Ti, C)-Fe<sub>2</sub>O<sub>3</sub>. Except the signals of Sn and Si from the FTO substrate and the C element from the adsorbed organics, only the signals of Fe and O elements can be detected. After adding glucose and TiCl<sub>3</sub> and PEC deposition of CoOOH, the obtained CoOOH/(Ti, C)-Fe<sub>2</sub>O<sub>3</sub> nanorods show obvious XPS signals of Ti (Fig. S6a), demonstrating the successful doping of Ti, and signals of Fe element origin from Fe<sub>2</sub>O<sub>3</sub> (Fig. S6b), signals of Co origin from CoOOH (Fig. S6c), signals of O origin from Fe<sub>2</sub>O<sub>3</sub> and CoOOH (Fig. S6d) [20,22,23,31].

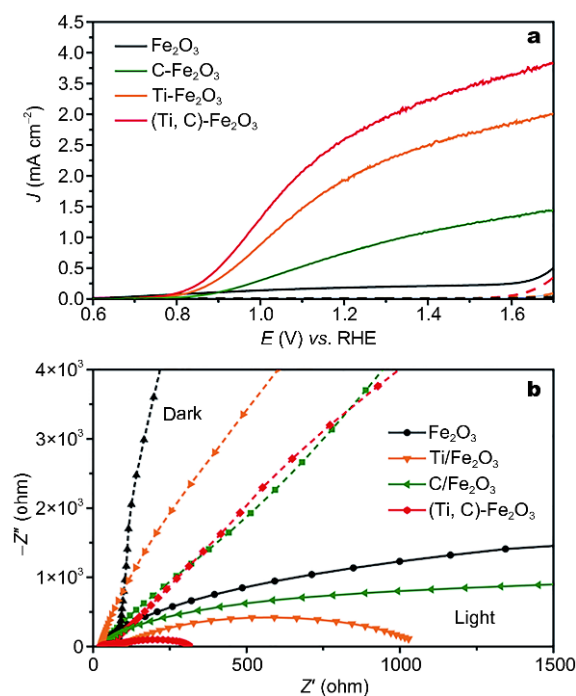
TEM and high resolution TEM (HRTEM) were carried out to present the structure and morphology of the CTCF. The diameter of Fe<sub>2</sub>O<sub>3</sub> nanorods is approximate 100 nm in agreement with the SEM result. The HRTEM



**Figure 2** (a) TEM and (b) HRTEM images of the CoOOH/(Ti, C)-Fe<sub>2</sub>O<sub>3</sub> nanorods. (c) HAADF-STEM image of the CoOOH/(Ti, C)-Fe<sub>2</sub>O<sub>3</sub> nanorods, and (d–h) the corresponding STEM-EDS elemental mapping images of Fe, O, C, Ti and Co, respectively.

image (Fig. 2b) shows the spacing lattice fringes is about 0.369 nm, corresponding to the (012) lattice planes of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, and in accordance with the XRD spectra. The results of TEM-energy dispersive spectrometer mapping (TEM-EDS) of CTFC nanorods reveal the homogeneous distribution of Fe, O, C, Ti and Co in the whole selection area (Fig. 2c–h).

The systematic electrochemical and PEC performance of both the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photoanode and the TCF photoanodes were characterized in (0.5 mol L<sup>-1</sup>), Na<sub>2</sub>SO<sub>3</sub> aqueous solution with NaOH solution (pH 14) as a hole scavenger, under AM 1.5G illumination. Fig. 3a compares the linear sweep voltammograms (LSV) curves of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photoanode, C-Fe<sub>2</sub>O<sub>3</sub> photoanode, Ti- $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photoanode and TCF photoanode. Even  $J_{\max}$  of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> is more than 15 mA cm<sup>-2</sup>, the photocurrent density of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> just only achieves 0.19 mA cm<sup>-2</sup> at +1.23 V vs. RHE, due to the poor electrical conductivity. However, after C and Ti doping, the TCF photoanode achieves a photocurrent density of 2.69 mA cm<sup>-2</sup> at +1.23 V vs. RHE, which is much higher than that of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photoanode (0.19 mA cm<sup>-2</sup>), C-Fe<sub>2</sub>O<sub>3</sub> photoanode (0.80 mA cm<sup>-2</sup>) and Ti-Fe<sub>2</sub>O<sub>3</sub> photoanode (2.07 mA cm<sup>-2</sup>), because of the enhanced electrical conductivity due to the C and Ti

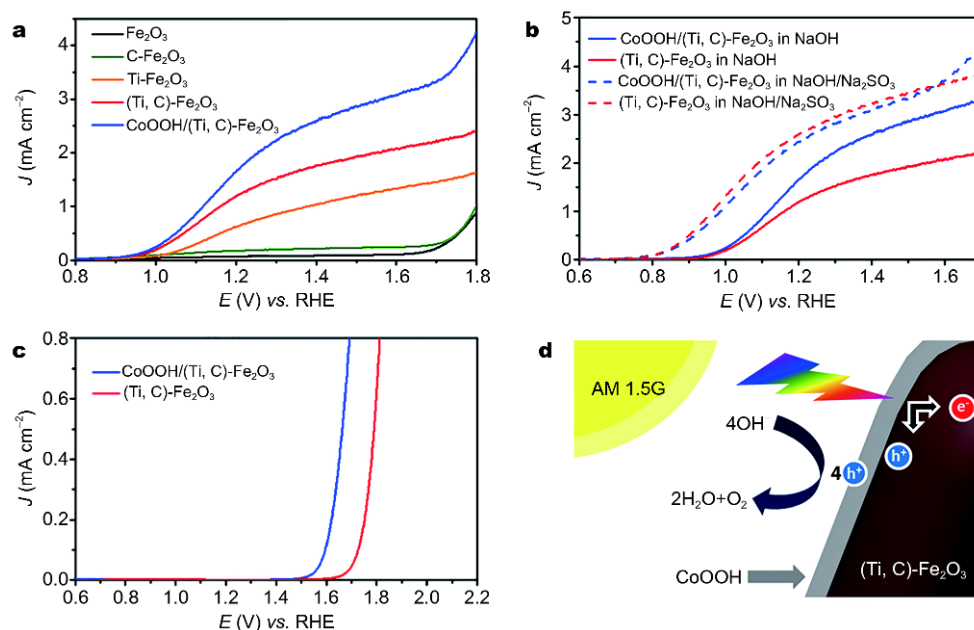


**Figure 3** (a) LSV curves recorded at a scan rate of 25 mV s<sup>-1</sup> under AM 1.5G irradiation in Na<sub>2</sub>SO<sub>3</sub> electrolyte with NaOH as a hole scavenger (pH 14); (b) EIS Nyquist plots of the as-prepared samples.

doping in  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (Fig. 3b). Electrochemical impedance spectroscopy (EIS) were carried out to further investigate the relationship between doping and electrical conductivity. As shown in Fig. 3b, the arc radius of CTF photoanode both in the dark and under light is smaller than those of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photoanode, C- $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photoanode, and Ti- $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photoanode in (0.5 mol L<sup>-1</sup>) Na<sub>2</sub>SO<sub>3</sub> aqueous solution with NaOH solution (pH 14) as a hole scavenger at 1.23 V vs. RHE. The results of EIS confirm the C and Ti doping can increase the electrical conductivity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. Increasing the electrical conductivity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> makes CTF photoanode achieve remarkable photocurrent density in solution with hole scavenger. Therefore, the CoOOH was used as OEC, which were deposited on the CTF photoanode surface by a PEC deposition method, to make maximum use of this photocurrent density of CTF in the solution without hole scavenger. In Fig. S7, the thickness of the CoOOH layer is about 5 nm and amorphous.

Fig. 4a compares the LSV curves of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photoanode, C-Fe<sub>2</sub>O<sub>3</sub> photoanode, Ti- $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photoanode, TCF photoanode, and CTFC photoanode at a scan rate of 25 mV s<sup>-1</sup> under the AM 1.5G irradiation. The CTFC photoanode achieves a photocurrent density of 1.85 mA cm<sup>-2</sup> at +1.23 V vs. RHE in the NaOH solution





**Figure 4** (a) LSV curves recorded at a scan rate of  $25 \text{ mV s}^{-1}$  under AM 1.5G irradiation in NaOH electrolyte (pH 14) with no hole scavenger and (b) with  $\text{Na}_2\text{SO}_3$  as a hole scavenger. (c) LSV curves recorded at a scan rate of  $25 \text{ mV s}^{-1}$  in NaOH electrolyte (pH 14) with no hole scavenger under dark condition. (d) The illustration of PEC water oxidation at the CTCF photoanode.

without hole scavenger (pH 14), which is much higher than that of the  $\alpha\text{-Fe}_2\text{O}_3$  photoanode ( $0.08 \text{ mA cm}^{-2}$ ),  $\text{C-Fe}_2\text{O}_3$  photoanode ( $0.18 \text{ mA cm}^{-2}$ ),  $\text{Ti-Fe}_2\text{O}_3$  photoanode ( $0.71 \text{ mA cm}^{-2}$ ), TCF photoanode ( $1.33 \text{ mA cm}^{-2}$ ). Due to the CoOOH OEC's effect, the CTCF photoanode achieves higher PEC water splitting performance than TCF photoanode, as illustrated in the LSV curves of TCF photoanode and CTCF photoanode in NaOH solution without hole scavenger and NaOH solution with  $\text{Na}_2\text{SO}_3$  hole scavenger (Fig. 4b).

Since the oxidation rate of holes ( $\Phi_{\text{OX}}$ ) on the electrolyte interface is very fast in the presence of  $\text{Na}_2\text{SO}_3$  as hole scavenger, the surface recombination of charges is eliminated and  $\Phi_{\text{OX}} \approx 100\%$ . As shown in Fig. 4b, when  $\text{Na}_2\text{SO}_3$  is added to the electrolyte, the photocurrents generated from CTCF photoanode and TCF photoanode under AM 1.5G illumination are  $2.57 \text{ mA cm}^{-2}$  and  $2.69 \text{ mA cm}^{-2}$  at  $1.23 \text{ V vs. RHE}$ , respectively. The  $\Phi_{\text{OX}}$  can be individually calculated using equation  $\Phi_{\text{OX}} = J_{\text{PEC}}/J_{\text{HS}}$ , and are plotted in Fig. S8. The  $\Phi_{\text{OX}}$  of CTCF photoanode (68.2%,  $+1.23 \text{ V vs. RHE}$ ) is significantly higher than that of TCF (46.1%) at  $+1.23 \text{ V vs. RHE}$ , which means that the improvement is from increasing holes OER capacity by CoOOH OEC. On the other hand, as shown in Fig. 4b, the CoOOH OEC not only can increase the OER capacity of photoanodes in the solution without

hole scavenger, but also can enhance the photocurrent density of photoanode in the solution with  $\text{Na}_2\text{SO}_3$  as hole scavenger, which means the CoOOH OEC will not reduce the charges separation of photoanodes.

Fig. 4c shows the LSV curves of CTCF photoanode and TCF photoanode in NaOH solution (pH 14) without hole scavenger and illumination. The LSV curve of the CTCF photoanode generates a current density of  $0.5 \text{ mA cm}^{-2}$  at a potential of  $+1.66 \text{ V vs. RHE}$ , which is much better than that of TCF photoanode ( $+1.79 \text{ V vs. RHE}$ ). These results verify that the CoOOH OEC can increase the OER capacity of photoanode in PEC water splitting.

The illustration of the PEC water oxidation at the CTCF photoanode is shown in Fig. 4d. When sun light radiates on the CTCF photoanode, the photo-generated electrons and holes are generated by  $(\text{Ti, C})\text{-Fe}_2\text{O}_3$ . In this process, due to the increased electrical conductivity of photoanode by C and Ti doping, the charges can transfer faster in photoanode. Then, photo-generated holes transfer to the CoOOH OEC layer by bias voltage and oxidize the  $\text{OH}^-$  of solution on CoOOH layer surface. Therefore, the C and Ti in this PEC system can increase the photo-generated charges transport, and CoOOH OEC layer can increase the OER capacity as an OEC material.

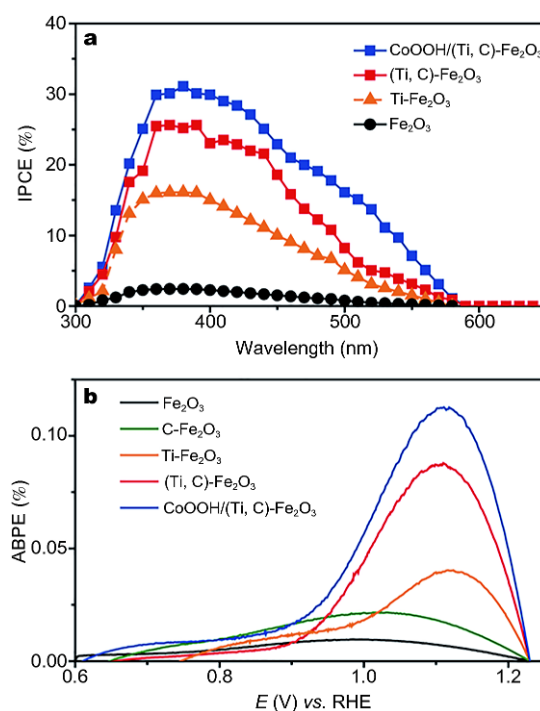
The IPCE were further performed to investigate the PEC performances of  $\alpha\text{-Fe}_2\text{O}_3$  photoanode,  $\text{Ti-Fe}_2\text{O}_3$

photoanode, TCF photoanode, and CTCF photoanode at +1.23 V vs. RHE in NaOH solution (pH 14). Fig. 5a shows the IPCE spectra of CTCF photoanode and TCF photoanode at +1.23 V vs. RHE, consistent with their current density-voltage ( $J$ - $V$ ) characteristics. The maximum IPCE value of CTCF photoanode reaches 31.2% at 380 nm, which is much higher than that of  $\alpha$ - $\text{Fe}_2\text{O}_3$  photoanode (2.2%),  $\text{Ti-Fe}_2\text{O}_3$  photoanode (16.1%) and TCF photoanode (25.7%). As shown in Fig. 5b, with the calculated ABPE as a function of external potential vs. RHE, the maximum efficiency for  $\alpha$ - $\text{Fe}_2\text{O}_3$  photoanode,  $\text{C-Fe}_2\text{O}_3$  photoanode,  $\text{Ti-Fe}_2\text{O}_3$  photoanode, TCF photoanode, and CTCF photoanode is 0.01% (0.99 V vs. RHE), 0.02% (1.01 V vs. RHE), 0.04% (1.12 V vs. RHE), 0.08% (1.08 V vs. RHE) and 0.11% (1.11 V vs. RHE), respectively. This result suggests that C and Ti doping and CoOOH OEC can increase PEC water splitting capacity of the  $\alpha$ - $\text{Fe}_2\text{O}_3$ -based photoanode.

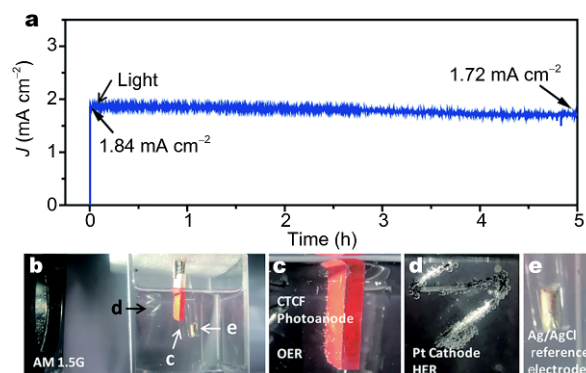
As shown in Fig. 6a and Movie S1, the CTCF photoanode exhibits excellent activity and operational stability for PEC water splitting in NaOH solution (pH 14) at 1.23 V vs. RHE under AM 1.5G irradiation. The chronoamperometry ( $i$ - $t$ ) curve of the CTCF photoanode was collected at 1.23 V vs. RHE in 5 h. The photocurrent of CTCF is initially about  $1.84 \text{ mA cm}^{-2}$  and decreases by  $0.12 \text{ mA cm}^{-2}$  within 5 h, demonstrating that the CTCF photoanode is very stable during the long time irradiation at extreme voltage in NaOH (pH 14). The SEM and XRD of the photoanode after stability test are shown in Fig. S9 and S10. And there is no change in morphology and structure after 5 h reaction. These results confirm the high stability of the CTCF photoanode. Fig. 6b is the photo image of CTCF photoanode PEC water splitting system, and Fig. 6c-e are photo images of different parts of this system. We also tested the hydrogen generation rate at  $1 \text{ mA cm}^{-2}$  (Fig. S11). We find the generation rate of  $\text{H}_2$  by our PEC system is about  $18.67 \mu\text{mol h}^{-1}$ , with Faradic efficiency close to 99.89%.

## CONCLUSIONS

In summary, the CTCF photoanode was successfully prepared by hydrothermal, calcination and PEC deposition methods. XRD, SEM, TEM and XPS results confirmed the Ti and C were successfully doped in the  $\text{Fe}_2\text{O}_3$ , and the layered CoOOH OEC was on the surface of (Ti, C)- $\text{Fe}_2\text{O}_3$ . The CTCF photoanode shows significantly PEC water splitting capacity under sunlight (AM 1.5G). The photocurrent density of CTCF photoanode is  $1.85 \text{ mA cm}^{-2}$  at +1.23 V vs. RHE, which is more than 20 times higher than that of pristine  $\alpha$ - $\text{Fe}_2\text{O}_3$  photoanode



**Figure 5** (a) IPCE and (b) ABPE spectra of the as-prepared photoanodes measured at bias +1.23 V vs. RHE.



**Figure 6** (a) Operational stability of the CTCF photoanode. Chronoamperometry ( $i$ - $t$ ) curve of CTCF collected at +1.23 V vs. RHE under AM 1.5G illumination in NaOH electrolyte (pH 14). Photo images of CTCF photoanode in PEC water splitting system (b), CTCF photoanode (c), Pt cathode (d) and Ag/AgCl reference electrode (e).

( $0.08 \text{ mA cm}^{-2}$ ). The IPCE, ABPE and  $\Phi_{\text{ox}}$  of the CTCF photoanode reach 31.2% at 380 nm (+1.23 V vs. RHE), 0.11% (+1.11 V vs. RHE), 68.2% (+1.23 V vs. RHE), respectively, which are much higher than those of pristine  $\alpha$ - $\text{Fe}_2\text{O}_3$  photoanode. The long time irradiation PEC water splitting of CTCF photoanode demonstrates its high stability at extreme voltage in NaOH (pH 14). These findings indicate the CTCF photoanode is a very pro-

mising material for PEC water splitting.

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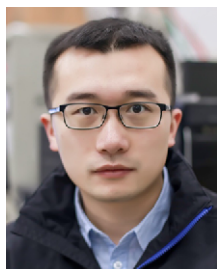
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**Conflict of interest** The authors declare no conflict of interest.

**Supplementary information** Supporting data are available in the online version of the paper.



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## 新型CoOOH/(Ti, C)-Fe<sub>2</sub>O<sub>3</sub>纳米棒光阳极制备及其光电解水性能研究

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**摘要** 本论文利用一种温和的方法合成了CoOOH/(Ti, C)-Fe<sub>2</sub>O<sub>3</sub> (CTCF)纳米棒光阳极, 并对其光电解水性能进行了研究. 在可见光照和1.0 V偏压(相对可逆氢电极)条件下, CTCF光阳极产生的光电流密度为1.85 mA cm<sup>-2</sup>, 远高于传统的 $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>光阳极的光电流密度. 同时, 该电极在强碱性电解液中(pH 14)可以保持较长时间的稳定性.